

High-Mobility Flexible Oxyselenide Thin-Film Transistors Prepared by a Solution-Assisted Method

Congcong Zhang, Jinxiong Wu,* Yuanwei Sun, Congwei Tan, Tianran Li, Teng Tu, Yichi Zhang, Yan Liang, Xuehan Zhou, Peng Gao, and Hailin Peng*



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ABSTRACT: Two-dimensional (2D) semiconductors hold great promise in flexible electronics because of their intrinsic flexibility and high electrical performance. However, the lack of facile synthetic and subsequent device fabrication approaches of high-mobility 2D semiconducting thin films still hinders their practical applications. Here, we developed a facile, rapid, and scalable solution-assisted method for the synthesis of a high-mobility semiconducting oxyselenide ($\text{Bi}_2\text{O}_2\text{Se}$) thin film by the selenization and decomposition of a precursor solution of $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$. Simply by changing the rotation speed in spin-coating of the precursor solution, the thicknesses of $\text{Bi}_2\text{O}_2\text{Se}$ thin films can be precisely controlled down to few atomic layers. The as-synthesized $\text{Bi}_2\text{O}_2\text{Se}$ thin film exhibited a high Hall mobility of $\sim 74 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ at room temperature, which is much superior to other 2D thin-film semiconductors such as transition metal dichalcogenides. Remarkably, flexible top-gated $\text{Bi}_2\text{O}_2\text{Se}$ transistors showed excellent electrical stability under repeated electrical measurements on flat and bent substrates. Furthermore, $\text{Bi}_2\text{O}_2\text{Se}$ transistor devices on muscovite substrates can be readily transferred onto flexible polyvinyl chloride (PVC) substrates with the help of thermal release tape. The integration of a high-mobility thin-film semiconductor, excellent stability, and easy transfer onto flexible substrates make $\text{Bi}_2\text{O}_2\text{Se}$ a competitive candidate for future flexible electronics.

High-mobility two-dimensional (2D) semiconductors, such as transition metal dichalcogenides (TMDCs), black phosphorus, InSe, and $\text{Bi}_2\text{O}_2\text{Se}$,^{1–5} have attracted considerable attention in post-silicon electronics. In contrast to typical bulk semiconductors, the intrinsic atomic thickness of 2D semiconductors brings a lot of unprecedented characteristics and unique applications.^{4,6–10} Except for much suppressed short channel effects in nanoscale transistors, the ultrathin body will also introduce excellent intrinsic flexibility and thus holds great promise for applications in flexible electronics such as flexible displays and integrated circuits.^{11–14} At present, despite the existence of a large family of 2D semiconductors, the lack of simple synthetic approaches of high-mobility 2D semiconducting thin films and compatible device fabrication for flexible electronics still hinders their practical applications.^{13,15–19} For example, continuous thin films of TMDCs (MoS_2) synthesized by chemical vapor deposition (CVD) or solution methods usually have relatively low mobility ($< 10 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$).^{20–22} Additionally, devices on flexible substrates usually exhibit much lower performance than those on rigid substrates. The performance degradation is mainly attributed to drawbacks of flexible substrates, such as substrate cleanliness, limited process temperature, and strain induced by roughness.^{23–25}

The pivotal part of the success of any materials in electronic applications is the development of a rapid, scalable, and facile process to synthesize high-quality thin films. In view of elemental components, $\text{Bi}_2\text{O}_2\text{Se}$ can derive from its parent compound of Bi_2O_3 by partially replacing O by Se.²⁶ Hence, it is reasonable to determine whether $\text{Bi}_2\text{O}_2\text{Se}$ can be obtained

by selenization of Bi_2O_3 , whose chemical reactions can be represented as follows: $\text{Bi}_2\text{O}_3 + \text{Se}_2(\text{g}) \rightarrow \text{Bi}_2\text{O}_2\text{Se} + \text{SeO}_2(\text{g})$.

As shown in Figure 1a,b, we developed a two-step method to synthesize a $\text{Bi}_2\text{O}_2\text{Se}$ thin film on flexible muscovite substrates, namely, depositing an ultrathin Bi_2O_3 film first, then selenizing Bi_2O_3 to get a $\text{Bi}_2\text{O}_2\text{Se}$ thin film. First, large-area ultrathin Bi_2O_3 films were synthesized by a solution-assisted and spin-coating method. $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$, a very cheap commercial reagent, was dissolved in ethylene glycol, then spin-coated onto a muscovite substrate at a certain rotation speed. Due to the excellent affinity between the solution and muscovite substrate, where the contact angle is as small as 6° (Figure S1), the $\text{Bi}(\text{NO}_3)_3$ solution spread out on the whole muscovite substrate with excellent uniformity. The nitrates on muscovite were then annealed in air to get uniform Bi_2O_3 thin films (Figures S2 and S3), whose thicknesses can be readily tailored by changing the rotation speed or the concentration of the Bi precursors. The decomposition reaction proceeded as follows: $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O} \rightarrow \text{Bi}_2\text{O}_3 + \text{NO}_x(\text{g}) + \text{O}_2(\text{g}) + \text{H}_2\text{O}$.

Figure 1c shows the typical microscopic image of as-synthesized $\text{Bi}_2\text{O}_2\text{Se}$ thin films on flexible muscovite substrates. As shown in Figure 1d and e, the identical contrast over a large scale indicates the excellent uniformity of the $\text{Bi}_2\text{O}_2\text{Se}$ thin film, exhibiting a very smooth surface with a surface roughness

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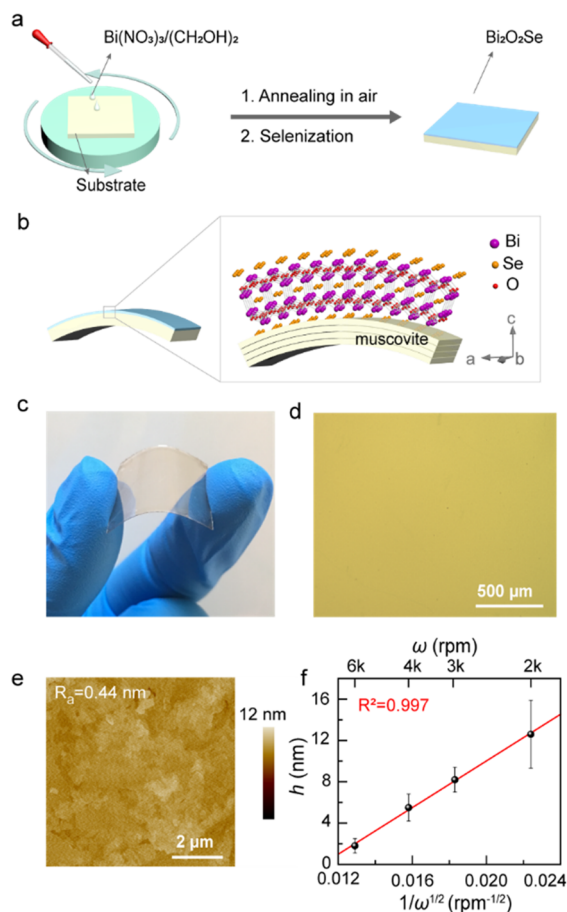


Figure 1. Schematic diagram and morphological characterization of a $\text{Bi}_2\text{O}_2\text{Se}$ thin film prepared by a solution-assisted method. (a) Preparation process of the $\text{Bi}_2\text{O}_2\text{Se}$ thin film, in which a $\text{Bi}(\text{NO}_3)_3/(\text{CH}_2\text{OH})_2$ solution was first spin-coated on a muscovite substrate, followed by annealing in air and another step of selenization to obtain a $\text{Bi}_2\text{O}_2\text{Se}$ thin film. (b) Diagram of a $\text{Bi}_2\text{O}_2\text{Se}$ thin film on a bendable muscovite substrate, which facilitates the fabrication of flexible electronics. (c) Photograph of an as-synthesized $\text{Bi}_2\text{O}_2\text{Se}$ thin film on a flexible muscovite substrate (2 cm \times 2 cm). (d) Typical optical microscope (OM) image of an as-synthesized $\text{Bi}_2\text{O}_2\text{Se}$ thin film with excellent uniformity. (e) Typical AFM image of a $\text{Bi}_2\text{O}_2\text{Se}$ thin film with a relatively small roughness ($R_a = 0.44$ nm). (f) Thickness (h) of the as-synthesized $\text{Bi}_2\text{O}_2\text{Se}$ thin film as a function of $\omega^{-1/2}$.

as small as 0.44 nm measured by atomic force microscopy (AFM). Additionally, spin coating is a facile way to change the thickness of the as-prepared thin film by changing the rotation speed, as the function that follows:

$$h \propto \left(\frac{\mu}{t\omega}\right)^{1/2}$$

where h is the thickness of the obtained thin film, μ stands for the viscosity of the solution, t represents the time of rotation, and ω is the rotation speed. As shown in Figures 1f and S4, the thickness (h) of the as-synthesized $\text{Bi}_2\text{O}_2\text{Se}$ thin film shows excellent linear dependence of $\omega^{-1/2}$, whose correlation coefficient is 0.997 (very close to 1). In other words, we can readily tailor the thickness of the as-synthesized $\text{Bi}_2\text{O}_2\text{Se}$ thin film even down to a few atomic layers, which is quite challenging for the traditional CVD method with very complicated elementary steps.

To investigate the crystal structure and domain size of the as-synthesized $\text{Bi}_2\text{O}_2\text{Se}$ thin film, we performed characterization by X-ray diffraction (XRD) and electron backscatter diffraction (EBSD). The as-synthesized $\text{Bi}_2\text{O}_2\text{Se}$ thin film can be readily transferred onto a glass slide for XRD characterization via a poly(methyl methacrylate) (PMMA)-assisted method. As indicated in Figure 2a, the $\text{Bi}_2\text{O}_2\text{Se}$ thin film

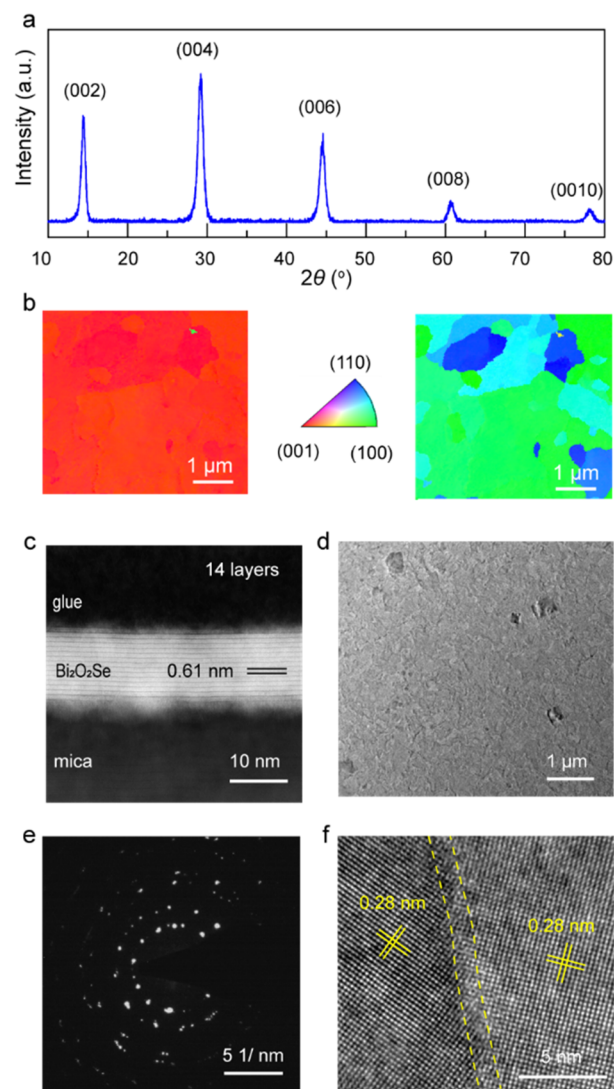


Figure 2. Structural characterization of a $\text{Bi}_2\text{O}_2\text{Se}$ thin film prepared by a solution-assisted method. (a) XRD pattern of a $\text{Bi}_2\text{O}_2\text{Se}$ thin film transferred onto a glass slide, showing only (001) indices. (b) EBSD patterns of a $\text{Bi}_2\text{O}_2\text{Se}$ thin film on muscovite, showing a domain size from hundreds of nanometers to several micrometers. (c) Cross-sectional STEM image of a $\text{Bi}_2\text{O}_2\text{Se}$ thin film on a muscovite substrate. (d) TEM image of a $\text{Bi}_2\text{O}_2\text{Se}$ thin film transferred onto a Cu grid, whose zone axis is [001]. (e) Typical SAED pattern of a $\text{Bi}_2\text{O}_2\text{Se}$ thin film with polycrystalline diffraction ring. (f) HRTEM of $\text{Bi}_2\text{O}_2\text{Se}$ domains merged together with different in-plane orientations.

showed only (001) indices, suggesting all the composed $\text{Bi}_2\text{O}_2\text{Se}$ domains adopted an “in-plane growth mode” (not vertical). The EBSD patterns of the $\text{Bi}_2\text{O}_2\text{Se}$ thin film on muscovite are consistent with the XRD results, showing a domain size from hundreds of nanometers to several micrometers (Figure 2b). As we know, electrons flowing

inside the layer usually dominate the electrical transport in a layered material. Therefore, the “in-plane growth mode” is typically preferable to keep high mobility for a layered thin film.

Cross-sectional scanning transmission electron microscopy (STEM) is a powerful tool for thin-film characterizations. As shown in Figure 2c, the as-synthesized Bi₂O₂Se thin film exhibits a typical layered structure, whose layer spacing of 0.61 nm is consistent with the theoretical value of the layer thickness in Bi₂O₂Se (0.608 nm). The film thickness is ~8.5 nm, which matches well with the AFM results (Figure S5). Additionally, we performed top-view transmission electron microscopy (TEM) characterizations of the as-synthesized Bi₂O₂Se thin film by transferring it onto a holey carbon supported Cu grid (Figure 2d–f). The typical selected area electron diffraction (SAED) pattern displayed several sets of diffraction patterns on a ring, suggesting the in-plane domain direction of Bi₂O₂Se is relatively random. To this end, a domain boundary will form when two domains with different crystal orientations merge together, which is clearly marked by the yellow dashed line in Figure 2f. The lattice spacing of 0.28 nm is identical to the theoretical value (0.27 nm) of lattice distance for the (110) plane in Bi₂O₂Se.

To evaluate the electrical properties of the as-synthesized Bi₂O₂Se thin film, we first patterned the Bi₂O₂Se continuous thin film (8 nm) into discrete 40 × 20 μm² sheets by a wet chemical etching method reported before²⁷ and then fabricated the sheets into standard Hall-bar devices (inset of Figure 3a).

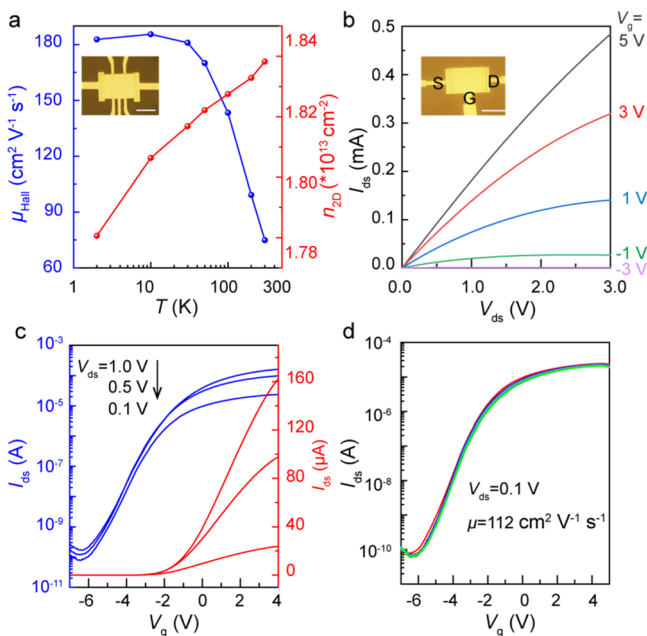


Figure 3. Electrical properties of a Bi₂O₂Se thin film prepared by a solution-assisted method. (a) Hall mobility (μ_{Hall}) and carrier concentration (n_{2D}) as a function of temperature in the Bi₂O₂Se thin film. Inset: OM image of a Hall-bar device fabricated on the 8 nm thick Bi₂O₂Se thin film. Scale bar, 20 μm. (b) Output curves obtained from an 8 nm thick Bi₂O₂Se transistor at room temperature. Inset: OM image of a top-gated device fabricated on a muscovite substrate, with Pd/Au electrodes and 20 nm HfO₂ serving as the top-gate dielectrics. (c) Transfer curves ($I_{\text{ds}}-V_{\text{g}}$) of the device with different source–drain voltages (1.0, 0.5, 0.1 V). (d) Repeating the transfer curve ($V_{\text{ds}} = 0.1$ V) tests 10 times, showing ignorable drift on threshold voltage and saturation current.

As shown in Figure 3a, the Hall mobility of an as-synthesized Bi₂O₂Se thin film can be as high as 74 cm² V⁻¹ s⁻¹ at room temperature and increases gradually upon cooling. It is worth noting that the room-temperature Hall mobility is much higher than other reported 2D semiconducting films and commercial InGaZnO thin films.^{25,28–31} Besides, the high-mobility 2D Bi₂O₂Se thin films were fabricated into top-gated field-effect transistors. As shown in Figure 3b, the $I_{\text{ds}}-V_{\text{ds}}$ shows a linear dependence at low drain–source biases, suggesting an ohmic contact was formed. Transfer curves ($I_{\text{ds}}-V_{\text{g}}$) of the device with different source–drain voltages (1.0, 0.5, 0.1 V) all exhibited a large on/off ratio of $>10^5$ (Figure 3c). By linear fitting of the transfer curves based on the equation $\mu_{\text{app}} = (L/W)(1/C_{\text{g}})(dI_{\text{ds}}/dV_{\text{g}})$, we can extract a two-probe apparent field-effect mobility as high as ~ 110 cm² V⁻¹ s⁻¹. Electrical stability is another key parameter that determines whether a device can be put into practical applications or not,^{32,33} since it is challenging to remove interfacial trap states, which will cause drain-induced barrier lowering (DIBL) under repeated transfer-curve measurements. Remarkably, the top-gated Bi₂O₂Se transistors showed ignorable drift on threshold voltage and saturation current (Figure 3d), which laid a solid foundation for practical applications that need excellent electrical stability, such as flexible electronics.

The feature of an ultrathin semiconducting film with intrinsic flexibility, excellent electrical stability, and high mobility in Bi₂O₂Se greatly facilitates its application in flexible electronics. As we know, because of the large surface roughness and incompatibility with the microfabrication process, it is very challenging to directly fabricate the devices on flexible polymer substrates, such as polyvinyl chloride (PVC). The preferable way is to fabricate the devices on a flat and rigid substrate first, then transfer the complete device onto other targeted flexible substrates, such as etching the sacrificial solid substrate.^{24,34–36}

Thanks to the layered structure, the muscovite substrates can be easily exfoliated into thin flakes at any thickness with excellent flexibility (Figure 4a) and stacks closely onto a flexible polymer substrate, such as PVC, to investigate the electrical performance of flexible transistors under repeated bending tests. First, we transfer the complete Bi₂O₂Se top-gated devices with ultrathin muscovite substrates underneath onto a thermal release tape; then the devices can be released onto the polymer substrates just by ~ 100 °C heating (Figure S6). This facile transferring process greatly facilitates the fabrication and electrical measurements of Bi₂O₂Se flexible transistors. Figure 4a demonstrates the diagram of a flexible Bi₂O₂Se transistor on muscovite/PVC substrates, and Figure 4b shows the devices with a curvature radius (R) of 4.0 mm. It is worth noting that drain current (I_{ds}) plotted against the gate voltage (V_{g}) shows a very slight change when the device is bent to 4.0, 5.3, and 7.8 mm (Figure 4c). Furthermore, the extracted field-effect mobilities of a flexible Bi₂O₂Se transistor bent at different curvature radii all have a similar value of ~ 100 cm² V⁻¹ s⁻¹ (Figure 4d). The transfer characteristics of flexible Bi₂O₂Se transistors and extracted mobilities also showed very slight changes after 0, 200, 400, ..., 1000 consecutive bending cycles at a 4.0 mm bending radius (Figure 4e,f). In a word, the Bi₂O₂Se thin films synthesized by a solution-assisted method showed excellent electrical stability when fabricated into flexible electronics.

In conclusion, high-mobility semiconducting Bi₂O₂Se thin films were successfully prepared by a rapid, scalable, and facile solution-assisted method. The precise control on the quantity

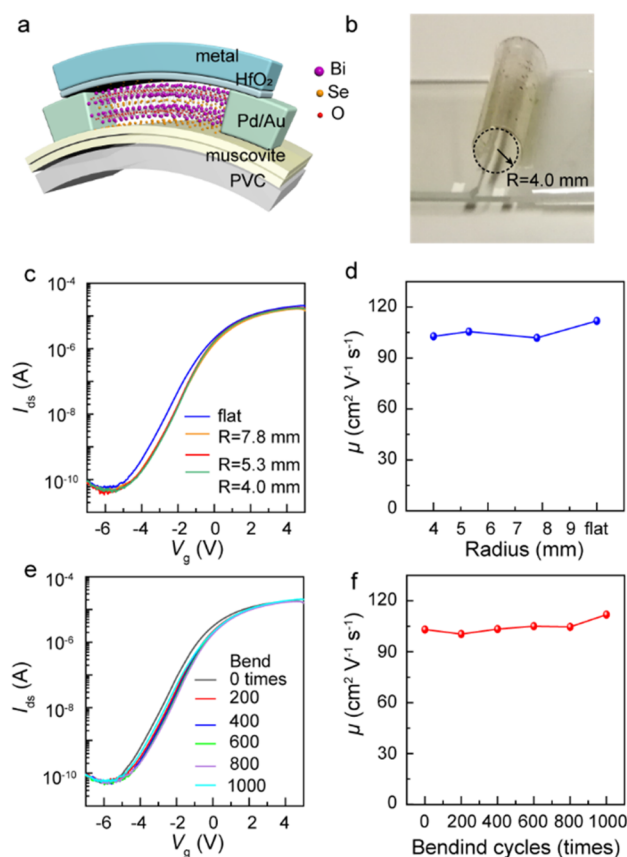


Figure 4. Electrical performance of a flexible $\text{Bi}_2\text{O}_2\text{Se}$ transistors under repeated bending tests. (a) Device configuration of a flexible $\text{Bi}_2\text{O}_2\text{Se}$ transistor on an ultrathin muscovite substrate with excellent flexibility. (b) Photograph of flexible $\text{Bi}_2\text{O}_2\text{Se}$ transistors on a muscovite/PVC substrate with a curvature radius (R) of 4.0 mm. (c) Drain current plotted against the gate voltage ($I_{\text{ds}}-V_{\text{g}}$), showing a very slight change when the device is bent to 4.0, 5.3, and 7.8 mm. (d) Extracted field-effect mobilities of a flexible $\text{Bi}_2\text{O}_2\text{Se}$ transistor bent at different curvature radii. (e) Transfer characteristics of flexible $\text{Bi}_2\text{O}_2\text{Se}$ transistors after 0, 200, 400, ..., 1000 consecutive bending cycles at a 4.0 mm bending radius. (f) Field-effect mobilities of the device as a function of bending cycles.

of Bi-containing precursors enables its facile thickness control of $\text{Bi}_2\text{O}_2\text{Se}$ thin film down to a few atomic layers just by changing the rotation speed. The integration of a high-mobility thin-film semiconductor, excellent stability, and easy transfer onto flexible substrates make $\text{Bi}_2\text{O}_2\text{Se}$ a competitive candidate for future flexible electronics.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/jacs.9b11668>.

Experimental details and supplementary figures (PDF)

■ AUTHOR INFORMATION

Corresponding Authors

Jinxiong Wu – Center for Nanochemistry, Beijing Science and Engineering Center for Nanocarbons, Beijing National Laboratory for Molecular Sciences, College of Chemistry and Molecular Engineering, Peking University, Beijing 100871, China; Tianjin Key Lab for Rare Earth Materials and

Applications, Center for Rare Earth and Inorganic Functional Materials, School of Materials Science and Engineering, National Institute for Advanced Materials, Nankai University, Tianjin 300350, China; Email: jxwu@nankai.edu.cn

Hailin Peng – Center for Nanochemistry, Beijing Science and Engineering Center for Nanocarbons, Beijing National Laboratory for Molecular Sciences, College of Chemistry and Molecular Engineering, Peking University, Beijing 100871, China; orcid.org/0000-0003-1569-0238; Email: hlpeng@pku.edu.cn

Authors

Congcong Zhang – Center for Nanochemistry, Beijing Science and Engineering Center for Nanocarbons, Beijing National Laboratory for Molecular Sciences, College of Chemistry and Molecular Engineering, Peking University, Beijing 100871, China

Yuanwei Sun – International Center for Quantum Materials, Electron Microscopy Laboratory, School of Physics, Peking University, Beijing 100871, China; Collaborative Innovation Center of Quantum Matter, Beijing 100871, China

Congwei Tan – Academy for Advanced Interdisciplinary Studies, Peking University, Beijing 100871, China

Tianran Li – Center for Nanochemistry, Beijing Science and Engineering Center for Nanocarbons, Beijing National Laboratory for Molecular Sciences, College of Chemistry and Molecular Engineering, Peking University, Beijing 100871, China

Teng Tu – Center for Nanochemistry, Beijing Science and Engineering Center for Nanocarbons, Beijing National Laboratory for Molecular Sciences, College of Chemistry and Molecular Engineering, Peking University, Beijing 100871, China

Yichi Zhang – Center for Nanochemistry, Beijing Science and Engineering Center for Nanocarbons, Beijing National Laboratory for Molecular Sciences, College of Chemistry and Molecular Engineering, Peking University, Beijing 100871, China

Yan Liang – Center for Nanochemistry, Beijing Science and Engineering Center for Nanocarbons, Beijing National Laboratory for Molecular Sciences, College of Chemistry and Molecular Engineering, Peking University, Beijing 100871, China

Xuehan Zhou – Center for Nanochemistry, Beijing Science and Engineering Center for Nanocarbons, Beijing National Laboratory for Molecular Sciences, College of Chemistry and Molecular Engineering, Peking University, Beijing 100871, China

Peng Gao – International Center for Quantum Materials, Electron Microscopy Laboratory, School of Physics, Peking University, Beijing 100871, China; Collaborative Innovation Center of Quantum Matter, Beijing 100871, China; orcid.org/0000-0003-0860-5525

Complete contact information is available at: <https://pubs.acs.org/doi/10.1021/jacs.9b11668>

Notes

The authors declare no competing financial interest.

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